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HIGH PRESSURE SYNTHESIS AND CRYSTAL DATA OF THE RARE EARTH ORTHOALUMINATES

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ABSTRACT

Single crystals of the rare earth orthoaluminates, REAlO₃, with RE = Dy - Lu, have been synthesized at high pressure in the presence of a flux. Lattice parameters for the orthorhombic members of the series, Sm-Lu, are reported. The a and c lattice parameters decrease smoothly in the direction Sm to Lu but the b parameter increases asymptotically to a maximum value at Lu. The anomalous behavior of the b parameter is similar to the variation of this parameter in the isostructural REFeO₃ series between La and Gd. For these latter compounds the variation has been explained in terms of a gradual change in the coordination number of the rare earth cation. A similar mechanism appears to be the cause in the case of the rare earth orthoaluminates.

Introduction

Unlike the ABO₃ perovskite-like compounds where A = rare earth and B = Fe, Ga, V, Cr, Rh, Sc and In, the rareearth orthoaluminates are not isostructural across the entireseries. From La through Nd the orthoaluminates have a trigonaldistortion of the ideal perovskite structure whereas from Smthrough Lu they have the orthorhombic distortion found in therare earth orthoferrites. The series is also unusual in thatthe synthesis of the members Dy-TmAlO₃ at atmospheric pressuregenerally yields a mixture of phases, namely the rare earthorthoaluminate plus the rare earth aluminum garnet(1). More

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recently Garton and Wanklyn have reported the synthesis of single crystals of Dy-YbAlO₂ using an equimolar mixture of oxides in PbO flux and a cooling rate from 1260°C of 50°C/hr (2). However, the Tm and Yb runs contained mixtures of orthoaluminate and garnet, while LuAlO3 could not be synthesized under these conditions.

From recent structural refinements of NdAlO3 and SmAlO₃, it has been shown that the coordination number of Nd^{3+} is twelve, and very nearly twelve for $\text{Sm}^{3+}(3)$. In contrast to this the coordination number of the rare earth ions in the (RE)FeO3 series is considerably less and varies from a maximum of approximately nine for LaFeO3 to a minimum of about eight for $LuFeO_3$ (4,5). It appears that in general the coordination numbers of the rare earth ions in the (RE)AlO3 series are larger than their iron counterparts. As a consequence of this it has been suggested that high pressures would favor the synthesis of the smaller rare earth orthoaluminates, especially $LuA10_{2}(3).$

This paper reports the synthesis of single crystals of REAlO3, where RE = Dy - Lu, via direct combination of equimolar mixtures of the oxides at high pressure in the presence of a flux. Also reported are the lattice parameters and X-ray powder patterns for the orthorhombic members of the series Sm-LuAlO2.

Experimental

High pressures were generated in a piston-cylinder device. An 0.5 inch diameter tungsten carbide piston was forced by means of an oil-driven hydraulic ram into a supported tungsten carbide pressure vessel. The pressure-transmitting medium in our pressure vessel was a talc cylinder with a carbon resistance heater. This apparatus has been more thoroughly described elsewhere (6). Equimolar mixtures of aluminum and rare earth oxides were ball-milled in ethanol for approximately 12 hours, filtered, and dried. The resulting powders were then mixed with crushed NaOH pellets in a mole weight ratio of roughly 2:1. Previously it was found that NaOH does not react to any appreciable extent with the rare earth oxides, rare earth

gallium garnets and Ga203(7), and for this reason was chosen as a flux for the orthoaluminates. Sample volumes of approximately 0.054 cc. each were packed into separate platinum containers. During all runs, two materials were pressed simultaneously at 32.5 kilobars and 1200°±10°C, for a period of four hours. After quenching, the apparatus was allowed to cool over night, the pressure released, and the subsequent recovery of the sample containers effected. Each platinum cell was then broken open and immersed in distilled water. The NaOH rapidly dissolved leaving small, rectangular parallelpiped-shaped, transparent crystals. The crystals of the Ho, Er, and Tm compounds were colored pale yellow, pale pink, and pale green respectively, while those of Dy, Yb and Lu aluminate were colorless. All the crystals were strongly birefringent.

X-ray powder films were taken of Eu-LuAlO3 crushed crystals with a Guinier camera using KCl as internal standard and CuKa radiation (Ka₁ = 1.5405 A and Ka₂ = 1.5443 A). None of the powder films showed any trace of garnet lines after 24 hrs. exposure. They were all indexed on the basis of the orthorhombic cell, space group Pbnm, with lattice constants given in Table 1.

TABLE 1

Orthorhombic Cell Data for the Series $SmAlO_3$ -LuAlO_3. The standard deviations are less than 10^{-4} , except for those of EuAlO_3 and GdAlO_3 which are twice as large.

	0	0	0	0
	a,(A)	b,(A)	c,(A)	Vol. A
SmAlO ₃ EuAlO3 GdAlO3 TbAlO3 DyAlO3 HoAlO3 ErAlO3 TmAlO3 YbAlO3 LuAlO3	5.2912 5.267 5.250 5.2317 5.2053 5.1811 5.1595 5.1435 5.1251 5.1012	5.2904 5.294 5.302 5.3097 5.3172 5.3229 5.3271 5.3277 5.3310 5.3310	7.4740 7.459 7.447 7.4196 7.3950 7.3741 7.3539 7.3335 7.3146 7.3000	209.22 208.0 207.3 206.11 204.68 203.37 202.12 200.96 199.85 198.55

The lattice parameters were refined from values of 2θ which were $< 2\theta(224)$, excluding overlapped lines. The lattice parameters for $SmAlO_3$ were determined by a simplified version

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of Bond's method (8,9). The powder patterns for Eu-LuAlO $_3$ are listed in Table 2.

Discussion

The lattice parameters of $SmAlO_3$ -LuAlO₃ are plotted against atomic number in Fig. 1. Atomic numbers rather than ionic radii were chosen as abscissae since the variation in coordination number across the series is not known. One can see that while the <u>a</u> and <u>c</u> parameters decrease in a nearly linear fashion from Sm-Lu, <u>b</u> increases asymptotically to the Lu value. The behavior of the <u>b</u> parameter resembles that of REGaO₃(10)compounds between Ce and Gd and the <u>b</u> parameter of the REFeO₃ series between La and Gd(11). However, in the case of the two latter series, the <u>b</u> values eventually decrease for smaller rare earth ions, but no decrease occurs for the REAIO₃ compounds.

The entire series of REFeO₃ structures have been refined from single crystal data in order to fully understand the mechanism underlying this anomalous behavior(4). It is now clear that the oxygen polyhedron around the rare earth ion is distorted and this distortion varies drastically across the series. The rare earth ion has approximately an eight fold



Lattice parameters a, b, and c versus atomic number for the rare earth orthoaluminates.

TABLE 2

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- -

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Eu

d_c

do

hkł

101

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Gd

de

do

Tb

dc

do

do

Dy	/	He	00	E1	r	T	n	Yt)	L.	L
	d _c	d _o	d _c	do	d _c	do	d _c	do	d _c	do	d,
						4.215 vw	4.211	4.203 vw	4.197	4.180 vw	4.181
m	3.720	3.714 s	3.713	3.709 m	3.706	3.700 ms	3.700	3.696 s	3.695	3.685 ms	3.686
m	3.698	3.688 ms	3.687	3.679 m	3.677	3.667 m	3.667	3.655 ms	3.657	3.651 wm	3.650
wm	3.323	3.316 m	3.316	3.310 m	3.310	3.303 m	3.304	3.298 m	3.298	3.291 m	3.290
wm	2.659	2.662 wm	2,661	2.663 wm	2.664	2.663 m	2.664	2.665 m	2.666	2.665 wm	2.666
vs	2.622	2.617 vs	2.616	2.611 s	2.610	2.603 s	2.605	2.598 vs	2.599	2.593 vs	2.594
wm	2.603	2.591 m	2.591	2.579 wm	2.580	2.572 m	2.572	2.562 m	2.563	2.550 wm	2.551
wm	2.502	2.503 wm	2.503	2.505 m	2.504	2.503 m	2.504	2.503 m	2.504	2.503 m	2.504
				2.214 w	2.214	2.208 w	2.209	2.202 w	5.505	2.194 w	2.194
m	2.159	2.159 m	2.158	2.158 wm	2.157	2.154 m	2.155	2.154 m	2.154	2.153 m	2.153
m	2.128	2.120 m	2,120	2.113 wm	2.112	2.104 m	2.105	2.098 ms	2.099	2.091 m	2.091
WW	2.055	2.050 w 1.992 vw	2.050	2.044 w	2.045	2.039 wm	2.040	2.035 m 1.986 vw	2.035	2.030 m	2.031
m	1.860	1.856 m	1.856	1.854 wm	1.853	1.850 m	1.850	1.848 m	1.847	1.843 m	1.843
m	1.849	1.843 m	1.844	1.839 wm	1.838	1.833 wm	1.833	1.828 m	1.829	1.824 wm	1.825
	1.807	1.805 w	1.806	1.803 vw	1.804	1.802 wm	1.801	1.799 wm	1.799	1.797 wm	1.797
wm	1.804	1.800 wm	1.800	1.797 m	1.797	1.795 Wm	1.794	1.791 m	1.791	1.787 m	1.787
vv	1.661	1.658 w	1.658	1.654 wm	1.655	1.652 W	1.652	1.649 w	1.649	1.645 wm	1.645
W	1.656	1.652 wm	1.651	1.647 wm	1.647	1.643 wm	1.643	1.639 wm	1.639	1.635 m	1.636
WW	1.649	1.643 VW	1.643	1 627 -	1 627	2 626 -	1.636	1.628 w	1.627	1.620 w	1.620
wm	1.636	1.636 wm	1.637	1.03/ m	1.031	T.030 m	1.632	1.636 m	1.636	1.635 m	1.636
	-					1.593 w	1.593	1.589 vw	1.588	AND A DESCRIPTION OF A	
wm	1.528	1.527 wm	1.528	1.527 wm	1.527	1.527 wm	1.526	1.526 wm	1.526	1.526 w	1.525
wm	1.518	1.515 wm	1.515			1.510 wm	1.510	1.508 wm	1.508	1.506 wm	1.506
	1.507	1.502 w	1.502	1.497 wm	1.497	1.493 w	1.493	1.489 w	1.489	1.485 wm	1.484
m	1.506	1.500 m	1.501	1.495 m	1.495	1.491 m	1.491	1.487 m	1.486	1.482 wm	1.481
		and the second s		1.478 w	1.478	1.476 wm	1.475	1.473 wm	1.472	1.470 w	1.469
W	1.387	1.386 wm	1.386	1.384 m	1.385	1.384 m	1.384	1.383 wm	1.383	1.382 wm	1.382
		the second second									

Guinier Diffraction Data for Some Rare Earth Orthoaluminates.

110	3.735 m	3.734	3.731 m 3.726 w	3.731	3.729 m 3.711 wm	3.727	3.722 m 3.699 m	3.720	3.714 s 3.688 ms	3.713	3.709 m 3.679 m	3.706	3.700 ms 3.667 m	3.700	3.696 s 3.655 ms	3.695
111	2 647 4	2 647	3.337 W	3.335	3.330 w	3.330	3.325 WTL	3.323	3.316 m	3.316	3.310 m	3.310	3.303 m	3.304	3.298 m	3.298
112	0.630	2.639	2.635 s	2.635	2.628 s	2.629	2.622 vs	2.622	2.617 vs	2.616	2.611 s	2.610	2.603 s	2.605	2.598 vs	2.599
200	2.039 5	2.634	2.626 W	2.625	2.616 w	2.616	2.603 wm	2.603	2.591 m	2.591	2.579 wm	2.580	2.572 m	2.572	2.562 m	2.563
021			2.499 vvw	2.497	2.500 VW	2.500	2.502 wm	2.502	2.503 wm	2.503	2.505 m	2.504	2.503 m	2.504	2.503 m	2.504
103											2.214 w	2.214	5.508 W	2.209	5.205 W	2.202
022	2.158 wm	2.159	2.159 W	2.159	2.159 W	2.159	2.158 m	2.159	2.159 m	2.158	2.158 wm	2.157	2.154 m	2.155	2.154 m	2.154
113	2.171 WI	2.171	2.147 W	2.145	2.130 W	2.130	2.120 m	2.120	2.120 1	2,120	2.113 Wm	2.045	2.104 m	2.105	2.090 ms	2.035
122					2.000 VW	2.001	2.0)) **	2.0))	1.992 VW	1.992	2:044 W	2.04)	2.039 WH	2.040	1.986 vw	1.986
220	1 965	1.867	1.865 wm	1.865	1.864 wm	1.863	1.860 m	1.860	1.856 m	1.856	1.854 wm	1.853	1.850 m	1.850	1.848 m	1.847
004	1.005 WH	1.865			1.855 w	1.855	1.849 m	1.849	1.843 m	1.844	1.839 wm	1.838	1.833 wm	1.833	1.828 m	1.829
023	1.815 W	1.812			1.808 w	1.810	1.805 wm	1.807	1.805 w	1.806	1.803 vw	1.804	1.802 wm	1.801	1.799 wm	1.799
221	, -	1.811		1 ((0	2.000 #	1.807	2 (62	1.804	1.800 wm	1.800	1.797 m	1.797	1.795 Wm	1.794	1.791 m	1.791
11/1	1.668 w	1.668	1.665 wm	1.666	1.005 VW	1.661	1.001 VW	1.656	1.050 W	1.651	1.054 Wm	1.000	1.052 W	1.6/12	1.649 W	1.649
310		1.000		1.000	1.657 VW	1.657	1.649 VV	1.649	1.643 VW	1.643	T.041 WII	1.041	T:042 MI	1.636	1.628 W	1.627
131			1.634 w	1.634	1.635 vw	1.635	1.636 wm	1.636	1.636 wm	1.637	1.637 m	1.637	1.636 m	1.632	1.636 m	1.636
311													1.593 w	1.593	1.589 vw	1.588
132			1.527 w	1.527	1.528 w	1.528	1.528 wm	1.528	1.527 wm	1.528	1.527 wm	1.527	1.527 wm	1.526	1.526 wm	1.526
024		1 500		3 530	1.520 w	1.521	1.518 wm	1.518	1.515 wm	1.515	1 107	2 1.07	1.510 wm	1.510	1.508 wm	1.508
204	1.522 w	1.522	1.518 wm	1.519	1.512 m	1.513	1.507 m	1.507	1.502 W	1.502	1.497 Wm	1.497	1.493 W	1.493	1.489 W	1.489
315		1.761		1.910		1.)13		1.900	1.900 m	1.901	1.478 w	1.478	1.476 wm	1.475	1.407 m	1.400
133							1.387 w	1.387	1.386 wm	1.386	1.384 m	1.385	1.384 m	1.384	1.383 wm	1.383
115															1.360 vw	1.360
041											1.310 w	1.310	1.310 vw	1.310	1.312 w	1.311
224	1.320 w	1.319			1.315 w	1.315	1.311 m	1.311	1.308 m	1.308	1.305 m	1.305	1.302 wm	1.302	1.300 wm	1.300
314									1 220 1	1 221					1.215 Wm	1.215
332							1.178 m	1.176	T.250 M	TICCT				1.170	1.168 wm	1.167
043													1.109 Wm	1.169		PE SERVICE
241													1.167 w	1.167		
116 420	1.179 w	1.180											1.160 wm	1.161	1.158 w	1.158
225													1.149 w	1.149	1.148 w	1.147
135														1.104		
422													1.105 wm	1.104		
200							•							1.104		

1.297 m 1.297

coordination between TbFeO₃ and NdFeO₃. At the extremities of the series this approximation breaks down. For LuFeO₃ the seventh and eighth nearest oxygen atoms are becoming second nearest neighbors, while for LaFeO₃ the ninth nearest oxygen is too close to be considered a next nearest-neighbor. This change in coordination number governs the behavior of the <u>b</u> parameter.

It seems likely that a similar mechanism applies in the case of the REALO₃ series. However, it is important to note that the orthorhombic series begins with SmALO₃ where the coordination number of Sm^{3+} is very nearly twelve, compared to eight for its iron counterpart. Also, the nonlinear variation of the <u>c</u> parameter and the significant change in slope of the b parameter between Sm and Tb in Fig. 1 suggest a rapid decrease in the coordination numbers of the rare earth ions. Between DyALO₃ and LuALO₃ the coordination number does not appear to decrease as drastically. However, without a detailed knowledge of the structure of at least several more REALO₃ members, it is difficult to ascertain how the rare earth polyhedron varies across the series.

Another interesting point is that starting with Ho one needs high pressures to synthesize single phase rare earth orthoaluminates. $LuAlO_3$ was easily formed at 32 kbar but no attempt was made to find the minimum pressure necessary for this synthesis. We suspect that 32 kbar exceeds the minimum considerably. It is a logical step to attempt to synthesize under pressure MALO₃, where M is of smaller ionic radius than Lu^{3+} . We believe In³⁺ and possibly Sc³⁺ are likely M-cations and expect to proceed with these experiments in the near future.

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